

## Determination of the critical nucleation frequency in undercooled metal melt \*

JIAN Zengyun (坚增运), CHANG Fang (常芳娥), YAN Wen (严文),

(Department of Materials Science and Engineering, Xi'an Institute of Technology, Xi'an 710032, China)

YANG Gencang (杨根仓) and ZHOU Yaohe (周尧和)

(State Key Laboratory of Solidification Processing, Northwestern Polytechnical University, Xi'an 710072, China)

Received April 12, 1999; revised May 10, 1999

**Abstract** By studying the effect of thermodynamic and kinetic factors on the nucleation frequency of undercooled metals, the expressions for heterogeneous and homogenous critical nucleation frequencies have been established. The results show that the homogenous critical nucleation frequency per unit volume is directly proportional to the ratio of the cooling rate to the volume of liquid metal. As the value of the equilibrium contact angle function  $f(\theta)$  of the most effective catalyst is constant, the heterogeneous critical nucleation frequency per unit area of the catalyst surface is directly proportional to the ratio of the cooling rate to the sum of the surface area of the most effective catalyst surface ( $R_c/VS_v$ ). When  $R_c/VS_v$  is constant, the heterogeneous critical nucleation frequency per unit area of the catalyst surface is inversely proportional to  $f(\theta)^{0.53}$ ; the critical nucleation frequency per unit continuous mass of the metal melt for both the homogenous and heterogeneous nucleation can be expressed in terms of a general formula. The critical nucleation frequency is slightly influenced by the nature of the metal. The obtained theoretical result agrees well with the homogenous critical nucleation frequency estimated by Turnbull.

**Keywords:** critical nucleation frequency, undercooling, cooling rate, catalyst.

The theories of homogenous nucleation and heterogeneous nucleation that described the relationship among nucleation frequency, nature of metal, nature of catalyst and undercooling in metal melt were derived by Turnbull in 1950s<sup>[1]</sup>. It is the critical nucleation frequency at which the first nucleus forms in the metal melt that has immediate meaning to the nucleation process of metal. Not only can the related physical parameters of metal and the characteristics of catalyst be determined, but also the nucleation process of metal can be controlled in terms of the critical nucleation frequency<sup>[2-7]</sup>. Turnbull was the first to estimate the critical nucleation frequency, who assessed that the homogenous critical nucleation frequency per droplet of diameter 50  $\mu\text{m}$  is  $10^{-1 \pm 1} \cdot \text{s}^{-1}$ <sup>[1]</sup>. Turnbull's method is imperfect in two aspects: first, lack of theoretical basis, the second is that the influence of technology factors on nucleation process was not considered. The accuracy of the result estimated by Turnbull has not been theoretically demonstrated yet. Another method of estimating the critical nucleation frequency is the calculating method<sup>[4-8]</sup>. Its basis lies on two formulas:  $I_v^*$  (homogenous critical nucleation frequency per unit volume)  $\times V$  (volume of liquid metal)  $\times t$  (nucleation time) = 1 or  $I_s^*$  (heterogeneous critical nucleation frequency per unit area of catalyst surface)  $\times S$  (total area of

\* Project supported by the National Natural Science Foundation of China (Grant No.59601011).

catalyst surface)  $\times t$  (nucleation time) = 1. The critical nucleation frequency of homogenous nucleation and heterogeneous nucleation can be calculated using these formulas. Willnecker and Herlach worked out the homogenous critical nucleation frequency per unit volume of Fe-based alloy with volume  $8 \times 10^{-8} \text{ m}^3$  is  $10^{5.48} \text{ s}^{-1} \cdot \text{m}^{-3}$  [5]. Although this method had theoretical basis, but the effect of undercooling (or, nucleation time) on nucleation frequency was not considered. In fact, nucleation frequency increases exponentially with the increase of undercooling<sup>[1]</sup>. The actual critical nucleation frequency, therefore, is much greater than the critical nucleation frequency calculated according to the above formulas.

The purpose of this paper is, through studying the influence of thermodynamic and kinetic factors on the critical nucleation frequency, to derive the theoretical relationship among the critical nucleation frequency, the physical character of metal, the nature of catalyst, the volume concerned and the cooling rate of liquid metal.

### 1 Critical nucleation frequency per unit volume of homogenous nucleation

Suppose that the volume and cooling rate of metal melt are  $V$  and  $R_c$ , then, the undercooling  $\Delta T^*$  at which the first nucleus forms in the melt can be calculated as:

$$1 = \int_0^{\Delta T^*} I_v \cdot V \cdot \frac{1}{R_c} \cdot d\Delta T, \quad (1)$$

where  $\Delta T$  is the undercooling of metal liquid,  $I_v$  is the homogenous nucleation frequency per unit volume.  $I_v$  can be expressed as<sup>[1]</sup>:

$$I_v = A_v \exp\left[-\frac{\Delta G_A}{k(T_m - \Delta T)}\right] \exp\left[-\frac{\alpha \sigma^3 T_m^2}{k \Delta H_v^2 (T_m - \Delta T) \Delta T^2}\right], \quad (2)$$

where  $A_v$  is estimated to be a constant ( $A_v \approx 10^{41 \pm 1} \text{ m}^{-3} \cdot \text{s}^{-1}$ ),  $\Delta G_A$  is the free energy of activation

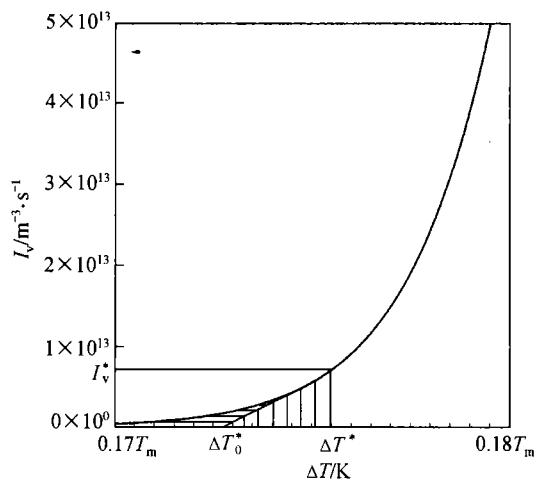


Fig. 1. Dependence of homogeneous nucleation frequency per unit volume on undercooling for copper.

for transporting an atom across the liquid-solid interface,  $k$  the Boltzmann constant,  $T_m$  the melting point of the metal,  $\alpha$  a factor determined by the shape of the nucleus (for spherical nucleus,  $\alpha = 16\pi/3$ ),  $\sigma$  the interface energy between solid and liquid, and  $\Delta H_v$  the latent heat of fusion per unit volume of metal.

The integral term in eq. (1) is related to the shaded area in fig. 1, but it can not be worked out directly. In order to calculate the shaded area, a tangent line to  $I_v$  curve at  $\Delta T^*$  in fig. 1 is drawn. Thus, the shaded area is divided into two parts. Because the left shaded area is much smaller than the shaded area on the right side of the tangent,

and because its effect on the undercooling is very small (the undercooling changes only 0.28% when the shaded area varies 50%), the whole shaded area can be replaced by the right shaded area approximately, i. e.

$$\begin{aligned} 1 &\approx \int_{\Delta T_0^*}^{\Delta T^*} \frac{dI_v}{d\Delta T} \Big|_{\Delta T = \Delta T^*} \cdot (\Delta T - \Delta T_0^*) \cdot V \cdot \frac{1}{R_c} \cdot d\Delta T \\ &= \frac{1}{2} V (\Delta T^* - \Delta T_0^*)^2 \frac{1}{R_c} \frac{dI_v}{d\Delta T} \Big|_{\Delta T = \Delta T^*}. \end{aligned} \quad (3)$$

From fig. 1, the following equation can be acquired:

$$\frac{dI_v}{d\Delta T} \Big|_{\Delta T = \Delta T^*} = \frac{I_v^*}{\Delta T^* - \Delta T_0^*}. \quad (4)$$

By differentiating eq. (2) with respect to  $\Delta T$ , the following equation can be obtained:

$$\frac{dI_v}{d\Delta T} \Big|_{\Delta T = \Delta T^*} = \left[ \frac{(2 - 3\xi^*)\psi}{(1 - \xi^*)^2 \xi^{*3} T_m} - \frac{\phi}{(1 - \xi^*)^2 T_m} \right] I_v^*, \quad (5)$$

where,  $\xi^* = \frac{\Delta T^*}{T_m},$  (6)

$$\psi = \frac{\alpha\sigma^3}{k\Delta H_v^2 T_m}, \quad (7)$$

$$\phi = \frac{\Delta G_A}{kT_m}, \quad (8)$$

and  $I_v^* = A_v \exp\left(-\frac{\phi}{1 - \xi^*}\right) \exp\left[-\frac{\psi}{(1 - \xi^*)\xi^{*2}}\right]$  (9)

is the critical nucleation frequency per unit volume of homogenous nucleation.

From eqs. (3), (4) and (5), the following equation can be derived:

$$I_v^* = \frac{2R_c}{T_m V} \frac{(2 - 3\xi^*)\psi - \phi\xi^{*3}}{(1 - \xi^*)^2 \xi^{*3}}. \quad (10)$$

By eliminating  $\xi^*$  from the simultaneous eqs. (9) and (10), a connotative relationship between  $I_v^*$  and  $V/R_c$  can be acquired. Substituting the related physical parameters of copper into eqs. (9) and (10), a curve representing the dependence of  $I_v^*$  on  $V/R_c$  can be obtained using Graphtool Software. Fig. 2 shows the dependence of  $\lg(I_v^*)$  on  $\lg(V/R_c)$  of copper. The result shows that  $\lg(I_v^*)$  decreases linearly with the increase of  $\lg(V/R_c)$ , which conforms the following rule:

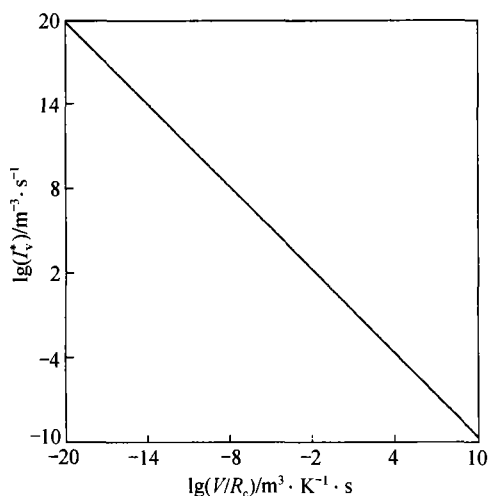


Fig. 2. Dependence of homogeneous critical nucleation frequency. Per unit volume  $I_v^*$  of copper on  $V/R_c$ .

ing the critical nucleation frequency per unit volume  $I_v^*$  into eqs. (9) and (10), the undercooling of metal at this solidification condition can also be calculated.

$$\lg(I_v^*) + \lg(V/R_c) = c_v, \quad (11)$$

namely 
$$I_v^* = 10^{c_v} \frac{R_c}{V}, \quad (12)$$

where  $c_v$  is a constant. Table 1 shows the calculated values of  $c_v$  and  $10^{c_v}$  for some metals. It is worth to point out that the values of  $c_v$  and  $10^{c_v}$  for different metals are almost constant, which shows that the critical nucleation frequency is influenced by the nature of metal only slightly.

The meaning of eq. (12) is that we can directly work out the critical nucleation frequency per unit volume  $I_v^*$  at a certain condition of solidification (e.g. when  $V$  and  $R_c$  are known). Furthermore, substituting

the critical nucleation frequency per unit volume  $I_v^*$  into eqs. (9) and (10), the undercooling of metal at this solidification condition can also be calculated.

Table 1 Values of  $c_v$  and  $10^{c_v}$  for some metals

	Metals					
	Cu	Au	Ag	Al	Fe	Ni
$c_v$	0.035	0.008	0.055	0.099	0.072	-0.080
$10^{c_v}$	1.08	1.02	1.13	1.26	1.18	0.83

## 2 The critical nucleation frequency of heterogeneous nucleation per unit area of catalyst surface

The heterogeneous nucleation process is influenced not only by the melt volume  $V$ , the melt cooling rate  $R_c$ , but also by the equilibrium contact angle function of effective catalyst  $f(\theta)$  ( $f(\theta) = (2 + \cos\theta)(1 - \cos\theta)^2/4$ ), and the surface area of the most effective catalyst per volume  $S_v$ . The heterogeneous nucleation frequency per unit area of catalyst surface  $I_s$  is obtained by substituting  $A_s$  ( $A_s = 10^{31 \pm 1} \cdot \text{m}^{-2} \cdot \text{s}^{-1}$  [11]) and  $\psi f(\theta)$  for  $A_v$  and  $\psi$  in eq. (2) respectively. Furthermore, substituting  $I_s$  and  $S_v V$  for  $I_v$  and  $V$  in eq. (1) respectively and treating them in the same way as homogeneous nucleation process, two equations corresponding to eqs. (9) and (10) can be obtained respectively, i.e.

$$I_s^* = A_s \exp\left(-\frac{\phi}{1 - \xi^*}\right) \exp\left[-\frac{\psi f(\theta)}{(1 - \xi^*) \xi^{*2}}\right], \quad (13)$$

$$I_s^* = \frac{2R_c}{T_m S_v V} \frac{(2 - 3\xi^*) \psi f(\theta) - \phi \xi^{*3}}{(1 - \xi^*)^2 \xi^{*3}}, \quad (14)$$

where  $I_s^*$  is the critical nucleation frequency per unit area of catalyst surface of heterogeneous nucleation.

Figure 3 shows the dependence of  $\lg(I_s^*)$  of copper on  $\lg(S_v V/R_c)$  at different values of  $f(\theta)$  by means of Graphtool Software according to eqs. (13) and (14). The result shows that the variation of  $I_s^*$  with  $S_v V/R_c$  has the same characteristics as that of  $I_v^*$  with  $V/R_c$ .  $I_s^*$  decreases with the increase of  $S_v V/R_c$ , and conforms to the following rule:

$$\lg(I_s^*) + \lg(S_v V/R_c) = c_s, \quad (15)$$

i.e.

$$I_s^* = 10^{c_s} \frac{R_c}{S_v V}, \quad (16)$$

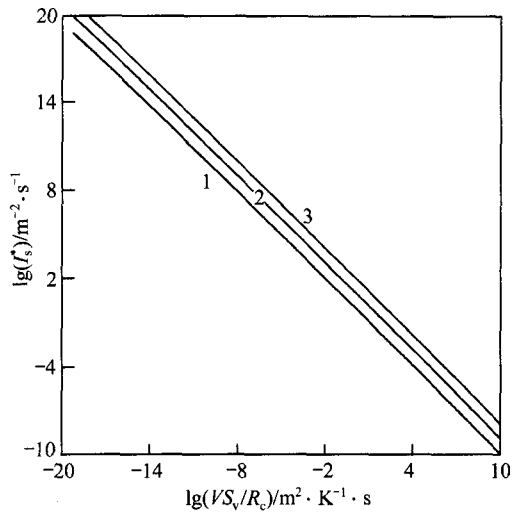


Fig. 3. The dependence of heterogeneous critical nucleation frequency. Per unit area  $I_s^*$  of copper on  $S_v V/R_c$ . 1,  $f(\theta) = 1$ ; 2,  $f(\theta) = 10^{-2}$ ; 3,  $f(\theta) = 10^{-4}$ .

where  $c_s$  is a constant. Table 2 shows the calculated values of  $c_s$  and  $10^c$  for some metals. It can be seen that the values of  $c_s$  and  $10^c$  are almost constant for different metals when  $f(\theta)$  is constant. But the values of  $c_s$  and  $10^c$  increase obviously with the decrease of  $f(\theta)$  for the same metal. Through regression analysis, the following regression equation is obtained:

$$c_s = c_s^0 - m \times \lg[f(\theta)], \quad (17)$$

where  $c_s^0$  is the value of  $c_s$  as  $f(\theta)$  is equal to 1,  $m$  is the regression coefficient.

Table 2 The values of  $c_s$ ,  $10^c$  and  $m$  for some metals

		Metals					
		Cu	Au	Ag	Al	Fe	Ni
$c_s$	$f(\theta) = 1$	0.08	0.05	0.10	0.18	0.13	-0.01
	$f(\theta) = 10^{-1}$	0.65	0.60	0.65	0.72	0.64	0.50
	$f(\theta) = 10^{-2}$	1.15	1.10	1.19	1.25	1.15	1.01
	$f(\theta) = 10^{-3}$	1.56	1.62	1.73	1.78	1.66	1.52
	$f(\theta) = 10^{-4}$	2.07	2.12	2.28	2.33	217	2.03
$10^c$	$f(\theta) = 1$	1.2	1.12	1.26	1.51	1.35	0.98
	$f(\theta) = 10^{-1}$	4.47	3.98	4.42	5.20	4.38	3.17
	$f(\theta) = 10^{-2}$	14.12	12.60	15.49	17.94	14.21	10.23
	$f(\theta) = 10^{-3}$	36.31	41.69	54.30	60.68	46.09	33.21
	$f(\theta) = 10^{-4}$	117.49	131.83	190.36	213.10	149.53	107.5
$m$		0.54	0.53	0.54	0.54	0.51	0.51

The values of  $m$  for some common metals are shown in table 2. It is worth to point out that the variations of  $m$  for different metals are insignificant. The slight discrepancies of values of  $c_s$ ,  $10^c$  and  $m$  for different metals illustrate that the effect of the nature of metal on the critical nucleation frequency is negligible.

Substituting eq. (17) into eq. (16), the general equation of heterogeneous critical nucleation frequency per unit area of catalyst surface can be obtained:

$$I_s^* = \frac{10^{\circ}}{f(\theta)^m} \frac{R_c}{S_v V}. \quad (18)$$

$S_v$  and  $f(\theta)$  are usually unknown. But, if the undercooling of liquid metal at a concrete solidification and melting condition are determined, the values of  $f(\theta)$  and  $S_v$  at this melting condition can be worked out according to eqs. (18), (14) and eqs. (13), (18)<sup>[9]</sup> respectively. When the values of  $f(\theta)$  and  $S_v$  are known, the critical nucleation frequency and undercooling of liquid metal at different  $R_c$  and  $V$  can be forecasted according to eqs. (18) and (13) respectively.

### 3 Discussion

If the nucleation frequency per unit continuous mass of metal melt is expressed as  $R_n$ , according to eqs. (12) and (18), the homogenous critical nucleation frequency per unit continuous mass of metal melt  $R_{nv}^*$  and the heterogeneous critical nucleation frequency per unit continuous mass of metal melt  $R_{ns}^*$  can be obtained respectively as follows:

$$R_{nv}^* + I_v^* V = 10^{\circ} R_c, \quad (19)$$

$$R_{ns}^* + I_s^* V S_v = \frac{10^{\circ}}{f(\theta)^m} R_c. \quad (20)$$

As mentioned above, the values of the constant term  $10^{\circ}$  in eq. (19) and  $10^{\circ}$  in eq. (20) vary in a very small range around 1, they are almost equal for different metals. If the values of  $10^{\circ}$  and  $10^{\circ}$  are assumed to be 1, and the value of  $m$  ( $m = 0.51-0.54$ ) is assumed to be 0.53, a general expression for the homogenous and heterogeneous critical nucleation frequency per unit continuous mass of metal melt can be obtained as:

$$R_n^* = \frac{R_c}{f(\theta)^{0.53}}, \quad (21)$$

where the unit of  $R_n^*$  is  $s^{-1}$  and it stands for the homogenous critical nucleation frequency per unit continuous mass of metal melt when  $f(\theta)$  is 1; and stands for the heterogeneous critical nucleation frequency per unit continuous mass of metal melt when  $f(\theta)$  is less than 1.

Equation (21) shows that the homogenous critical nucleation frequency per unit continuous mass of metal melt depends only on the cooling rate of liquid metal, but the heterogeneous critical nucleation frequency per unit continuous mass of metal melt depends not only on the cooling rate of liquid metal, but also on the contact angle of catalyst. Therefore, the homogenous critical nucleation frequency per unit continuous mass of metal melt can be calculated only if the cooling rate of liquid metal is known. But if we want to determine the heterogeneous critical nucleation frequency per unit continuous mass of metal melt, not only the cooling rate of liquid metal, but also the contact angle of catalyst must be known.

Substituting the experimental parameters ( $R_c = 0.5 \text{ Ks}^{-1}$ ,  $V = 6.5 \times 10^{-14} \text{ m}^3$ ) used by Turnbull into eq. (21), the homogenous critical nucleation frequency per unit continuous mass of metal melt  $R_{nv}^*$  can be calculated as  $10^{-0.3} \cdot \text{s}^{-1}$ , i.e.  $I_v^*$  is  $10^{12.9} \text{ m}^{-3} \cdot \text{s}^{-1}$ . The calculated result agrees well with the result ( $10^{-1 \pm 1} \cdot \text{s}^{-1}$ ) estimated by Turnbull. Substituting the experimental parameters ( $R_c = 8 \text{ Ks}^{-1}$ ,  $V = 8 \times 10^{-8} \text{ m}^3$ ) used by Willnecker into eq. (21), the homogenous critical nucleation frequency per unit continuous mass of metal  $R_{nv}^*$  and per unit volume  $I_v^*$  can be calculated as  $10^{0.90} \cdot \text{s}^{-1}$  and  $10^8 \text{ m}^{-3} \cdot \text{s}^{-1}$  respectively. The obtained  $I_v^*$  is almost  $10^3$  times higher than the result ( $10^{5.48} \cdot \text{s}^{-1} \cdot \text{m}^{-3}$ ) estimated by Willnecker et al.<sup>[5]</sup>. As stated above, this difference is caused by that the variation of nucleation frequency with undercooling was not considered in the formulas<sup>[4-8]</sup> used by Willnecker et al.

#### 4 Conclusion

(i) The homogenous critical nucleation frequency per unit volume  $I_v^*$  is directly proportional to the ratio of cooling rate to volume of liquid metal; the heterogeneous critical nucleation frequency per unit area of catalyst surface  $I_s^*$  is in direct proportion to the ratio of the cooling rate to the sum of the area of the most effective catalyst surface in the liquid metal ( $R_c/VS_v$ ) as  $f(\theta)$  is constant; when  $R_c/VS_v$  is constant, the critical nucleation frequency of heterogeneous nucleation per unit area of catalyst surface  $I_s^*$  is in inverse proportion to  $f(\theta)^{0.53}$ .

(ii) The critical nucleation frequency per unit continuous mass of metal  $R_n^*$  is influenced by the cooling rate of liquid metal and the equilibrium contact angle of the catalyst. It increases with the increasing cooling rate and decreases with the increase of equilibrium contact angle. The homogenous and heterogeneous critical nucleation frequency per unit continuous mass of metal melt can be expressed by a general equation:  $R_n^* = R_c \cdot f(\theta)^{-0.53}$ .

(iii) The critical nucleation frequency is slightly influenced by the nature of metal.

(iv) The obtained theoretical result agrees well with the homogenous critical nucleation frequency estimated by Turnbull.

#### References

- 1 Turnbull, D., Formation of crystal nuclei in liquid metals, *J. Appl. Phys.*, 1950, 21: 1 022.
- 2 Cantor, B., Doherty, R. D., Heterogeneous nucleation in solidifying alloys, *Acta metall.*, 1979, 27: 33.
- 3 Glicksman, M. E., Childs, W. J., Nucleation catalysis in supercooled liquid tin, *Acta metall.*, 1962, 10: 925.
- 4 Mueller, B. A., Perepezko, J. H., The undercooling of aluminum, *Metall. Trans. A*, 1987, 18: 1 143.
- 5 Willnecker, R., Herlach, D. M., Feuerbacher Containerless undercooling of bulk Fe-Ni melts, *Appl. Phys. Lett.*, 1984, 45: 615.
- 6 Willnecker, R., Herlach, D. M., Feuerbacher Nucleation in bulk undercooled of bulk nickel-base alloys, *Mater. Sci. Eng.*, 1988, 98: 85.
- 7 Wood, R. F., Lowndes, D. H., Narayan, J., Bulk nucleation and amorphous phase formation in highly undercooled molten silicon, *Appl. Phys. Lett.*, 1984, 44: 770.
- 8 Devaud, G., Turnbull, D., Undercooling of molten silicon, *Appl. Phys. Lett.*, 1985, 46: 884.
- 9 Jian, J. Z., Chang, F. E., Yan, W. et al., Determination of the catalyst in undercooled metal melt. *Academic Periodical Abstracts of China* (in Chinese), 1999, 5: 234.